

are taken in powder form and done using XRD-6100 (Shimadzu) diffractometer with Copper K_{α} radiation. The infrared spectra were recorded in the wave number range 400-2000 cm^{-1} using IR Affinity-1s (Shimadzu) spectrometer using KBr pellet technique at room temperature.

3. Results and Discussion

3.1 Physical Parameters

Using conventional formulae [13], density (ρ) of the synthesized glass samples, total molecular weight M of glass samples, various physical parameters such as transition metal ion concentration (Manganese ion), mean separation (r_i), polaron radius (r_p) and Optical basicity (Λ_{th}) values are determined.

3.1.1 Optical Basicity

The optical basicity of an oxide glasses can be conveniently measured in terms of the ability of the glass to donate a negative charge to an acidic probe ion [14-17]. Duffer and

Ingram [18] reported that the optical basicity can be predicted from the composition of the glass and the basicity moderating parameters of various cations present. The theoretical values of optical basicity (Λ_{th}) of the SLCB glasses can be estimated using the formula,

$$(\Lambda_{th}) = \frac{\sum_{i=1}^n \frac{Z_i r_i}{2\gamma_i}}{\quad} \quad (1)$$

where 'n' is the total number of cations present. Z_i is the 'i' is the oxidation number of the i^{th} cation to the number of oxides present and ' γ_i ' is the basicity moderating parameter of the i^{th} cation. The basicity moderating parameter ' γ_i ' can be calculated from the following equation

$$\gamma_i = 1.36(x_i - 0.26) \quad (2)$$

where x_i is the Pauling electro negativity of the cation [19]. The theoretical optical basicity values computed for the glasses in the present study are given in **Table 2**.

Table 2: Physical parameters of Manganese ions doped in SLCB glasses

Physical Parameters (Units)	Glass System			
	SLCB ₀	SLCB ₁	SLCB ₂	SLCB ₃
Average molecular weight(M)(g/mol)	67.69	64.01	67.70	71.39
Density (gr/cc)	2.65	2.79	2.67	2.54
Transition metal ion (N_i) Concentration (10^{+19} ions/cc) (± 0.005)	2.63	2.37	2.14
Inter ionic distance (r_i)(A^0)(± 0.005)	33.61	34.78	35.97
Polaron radius (r_p)(A^0)(± 0.005)	13.54	14.01	14.49
Optical basicity (Λ_{th})	0.43	0.43	0.43	0.43

It is observed that for the Λ_{th} values gradually increases from SLCB₁ to SLCB₃.

3.2 X-ray Diffraction Study (XRD)

The X-ray diffraction spectra have not shown any sharp peaks, indicating that the samples prepared are amorphous in nature are shown in **Figure 1**.

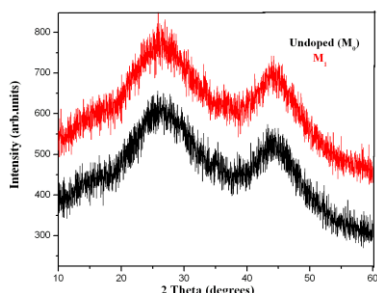


Figure 1: Powder X-ray diffraction patterns of undoped and Manganese ions doped SLCB glasses

3.6 FT-IR Studies

The FT-IR transmission spectrum is shown in Figure 2.

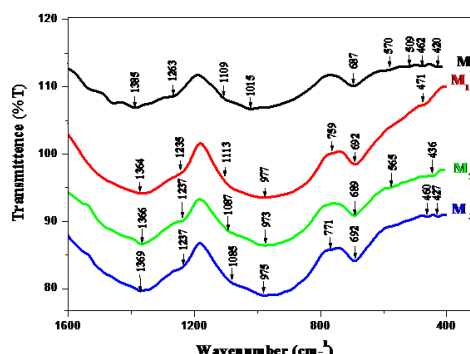


Figure 2: FT-IR spectra of Manganese ions doped SLCB glasses

The assignments of various absorption bands are given in **Table 3**.

The IR spectra have been divided into three spectral regions. The first group of band is reported to occur from 1200-

1600 cm^{-1} which arise due to the asymmetric stretching vibrations triangular BO_3 units. The second group of bands region is from 800-1200 cm^{-1} which was assigned to the stretching vibrations of tetrahedral BO_4 units. The third group observed around 700 cm^{-1} and was attributed to bending of B-

O-B linkages in the networks. The band at 900 cm^{-1} was a characteristic of glass with high boric acid content i.e. due to boroxil rings [20].

Table 3: Assignments of absorption bands in the FTIR spectra of Manganese ions doped in SLCB glasses

M_0	M_1	M_2	M_3	Assignment
420	-	436	427	Ca-O stretching mode of BO_3 units.
462	471	-	460	Ca-O ₆ stretching mode of BO_3 units.
509	-	-	-	Ca-O stretching mode of Borate units.
570	-	565	-	Ca-O stretching mode of Borate units.
687	692	689	692	B-O-B bending vibrations.
-	977	973	975	Boroxil ring vibrations.
1015	-	-	-	B-O-B stretching vibrations of BO_4 units.
1109	1113	1087	1085	Pyro-borate units, meta borate units.
1263	1235	1237	1237	Stretching vibrations B-O bonds in BO_3 units from pyro-ortho borate group's
1385	1364	1366	1369	Stretching vibrations of B (III)-O-B (IV) units.

The FT-IR spectrum of the undoped glass system contains three major broad bands with some shoulders in the wavenumber range of 400 cm^{-1} - 460 cm^{-1} , 500 cm^{-1} , 600 cm^{-1} . These bands are the characteristics band vibrations of the calcium-oxygen. When Li_2O is incorporated in the glass, some of borons become tetrahedrally coordinated. The spectrum of fused B_2O_3 consists of a completely continuous triangle like coordinated network and also contains some BO_4 tetrahedral coordination. The band which lies in the region 800 cm^{-1} – 1200 cm^{-1} splits into two broad bands whose intensity increases with increase in mol% of SrO and decrease of mol% of Li_2O content. In the undoped glass with x=5 mol% SrO and 15 mol% Li_2O a shift in vibrational band at 1385 cm^{-1} is noticed with feeble increase in intensity. When x=10 mol% SrO and Li_2O , Li_2O causes a shift in the vibrational band from 1364 cm^{-1} to 1366 cm^{-1} with a noticeable increase in intensity of these peaks. Another band at 1235 cm^{-1} with x=10 mol% SrO and x=10 mol% Li_2O is assigned to stretching vibrations of B-O bonds in BO_3 units from pyro-ortho borate groups. The bands 1109 cm^{-1} , 1113 cm^{-1} , 1087 cm^{-1} and 1085 cm^{-1} are assigned B-O stretching vibrations of tetrahedral BO_4 units in tri-borate, tetra borate and penta borate groups [21]. The bands at 977 cm^{-1} , 973 cm^{-1} , and 975 cm^{-1} are assigned to Boroxil ring vibrations. The bands 687 cm^{-1} , 682 cm^{-1} , 689 cm^{-1} and 692 cm^{-1} corresponds to B-O-B bending [22]. The bands 570 cm^{-1} , 565 cm^{-1} , 509 cm^{-1} are due to stretching vibrations of CaO stretching mode of Borate units [23]. The absorption bands 462 cm^{-1} , 471 cm^{-1} , 460 cm^{-1} and 420 cm^{-1} , 436 cm^{-1} , 427 cm^{-1} corresponds to Ca-O₆ stretching mode of BO_3 units [24]. It is observed that on increasing of SrO content, the frequency bands shifts from higher to lower wavenumber, which suggests the formation of non bridging oxygen (NBO'S) i.e. conversion of BO_3 to BO_4 structural units. The formation of NBO'S indicates that the addition of Lithium oxide (Li_2O) in SLCB glasses act as a modifier.

4. Conclusion

From the physical and optical characterization of 0.1 mol% of Manganese ions doped xSrO + (20x) Li_2O + (10-y) CaO + 70 B_2O_3 + y glasses with ($5 \leq x \leq 15\text{mol}\%$) the following conclusions are drawn.

- The density values are found to decrease with increase of SrO mol% or with decrease of Li_2O mol%. The density of undoped glass and the glass doped with Manganese oxide with 10 mol% of SrO and Li_2O are observed to be nearly equal.
- Powder XRD diffractograms confirm the glassy nature of the glasses under investigation.
- when x=5mol% of SrO and x=10 mol% Li_2O from the FT-IR spectra, the glass matrix is stable.

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